

SILVER METALLIZATION FOR ULSI APPLICATIONS

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Outline

- ◆ Introduction

- ◆ Results

 - PHASE I:* Blanket Thin Ag Films

 - PHASE II:* Patterned Ag Structures

- ◆ Summary

National Science Foundation (L. Hess)

Center for Low Power Electronics and member companies

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Technology Trend : Continuous Device Scaling

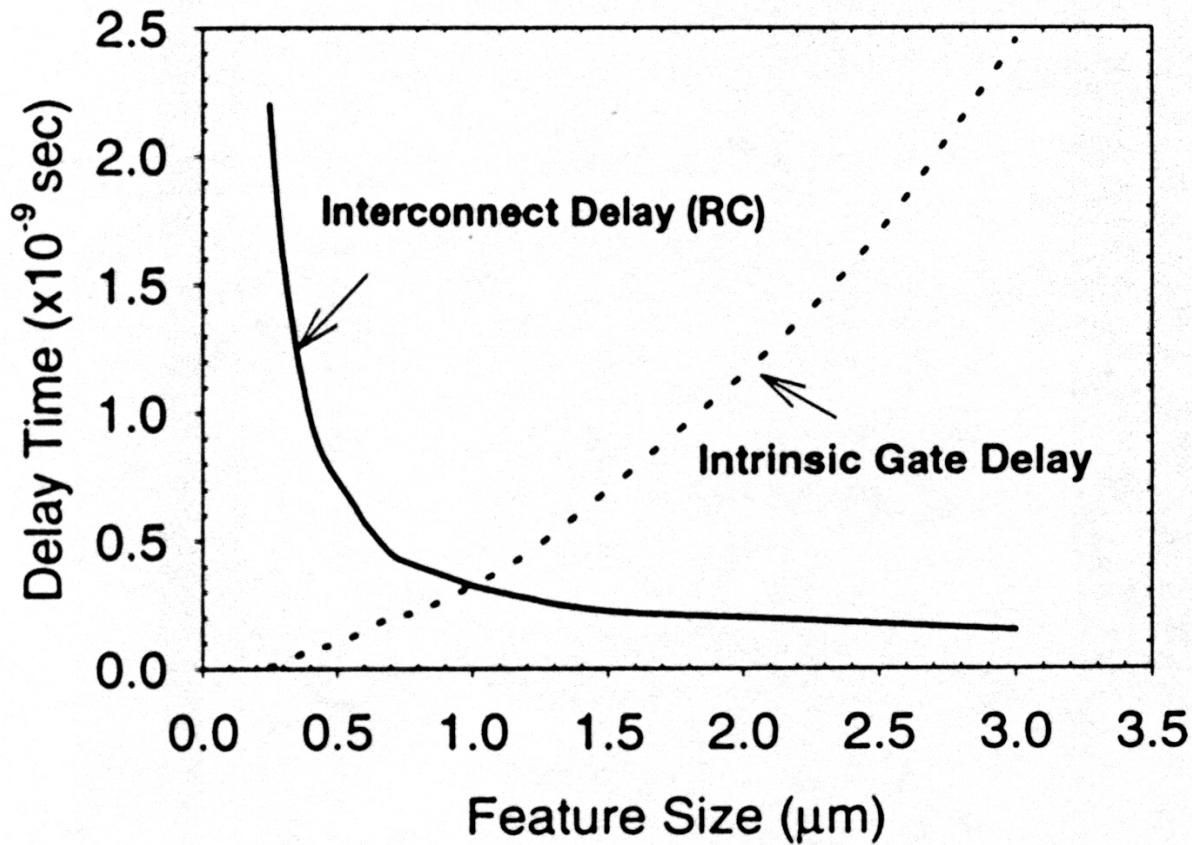
◆ BENEFITS

- *Faster circuit speed*
- *Higher functional density*

◆ ISSUES

- *Propagation delay*
- *Power dissipation*
- *Cross talk noise -*

Interconnect RC Delay: A Performance-Limiting Factor for Submicron Integrated Circuits



(S.-P. Jeng *et al.*, *Advanced Metallization for Devices and Circuits – Science, Technology, and Manufacturability*, 1994.)

Conventional Al Metallization

- ◆ Relatively high resistivity
- ◆ Electromigration (EM) failure

— *Questionable for ULSI*

Approach to Reduce Interconnect RC Delay

◆ Low- ρ metals to replace Al(Cu)

Al(Cu)	3.1 $\mu\Omega \cdot \text{cm}$
Au	2.4 $\mu\Omega \cdot \text{cm}$
Cu	1.7 $\mu\Omega \cdot \text{cm}$
<i>Ag</i>	<i>1.58 $\mu\Omega \cdot \text{cm}$</i>

◆ Low- k dielectrics to replace SiO₂

SiO ₂	$k = 4.0-4.5$
Low- k dielectrics	$k < 3$

Ag Metallization

◆ Advantages

- *Lower resistivity than Al and Cu (~ 40% and 5%)*
- *Higher EM resistance than Al*

◆ Disadvantages

- *Poor adhesion to SiO_2*
- *Rapid diffusion in Si*
- *Corrosion in S or Cl ambient*
- *Agglomeration*

Challenges of Ag Metallization

- ◆ **Address issues:**
 - *Diffusion*
 - *Corrosion*
 - *Agglomeration*
 - *Adhesion*

- ◆ **Improve EM resistance**

- ◆ **Produce Ag pattern**

Project Objective:

Demonstrate Ag as an interconnect material

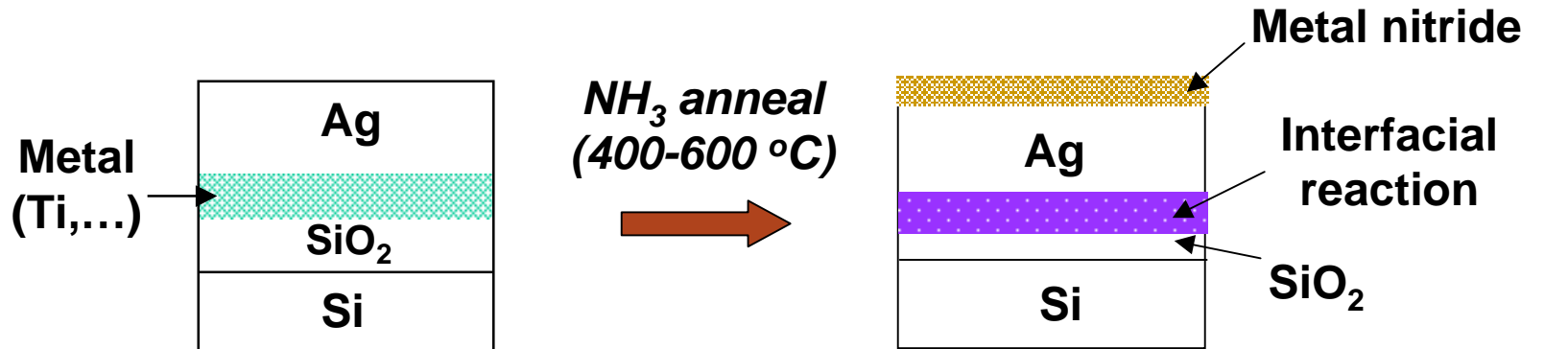
- ◆ ***Overcome Ag's critical issues***
- ◆ ***Develop Ag reactive ion etch***
- ◆ ***Optimize EM performance***

Strategy - Encapsulation Process

***PHASE I:* Blanket Thin Ag Films**

***PHASE II:* Patterned Ag Structures**

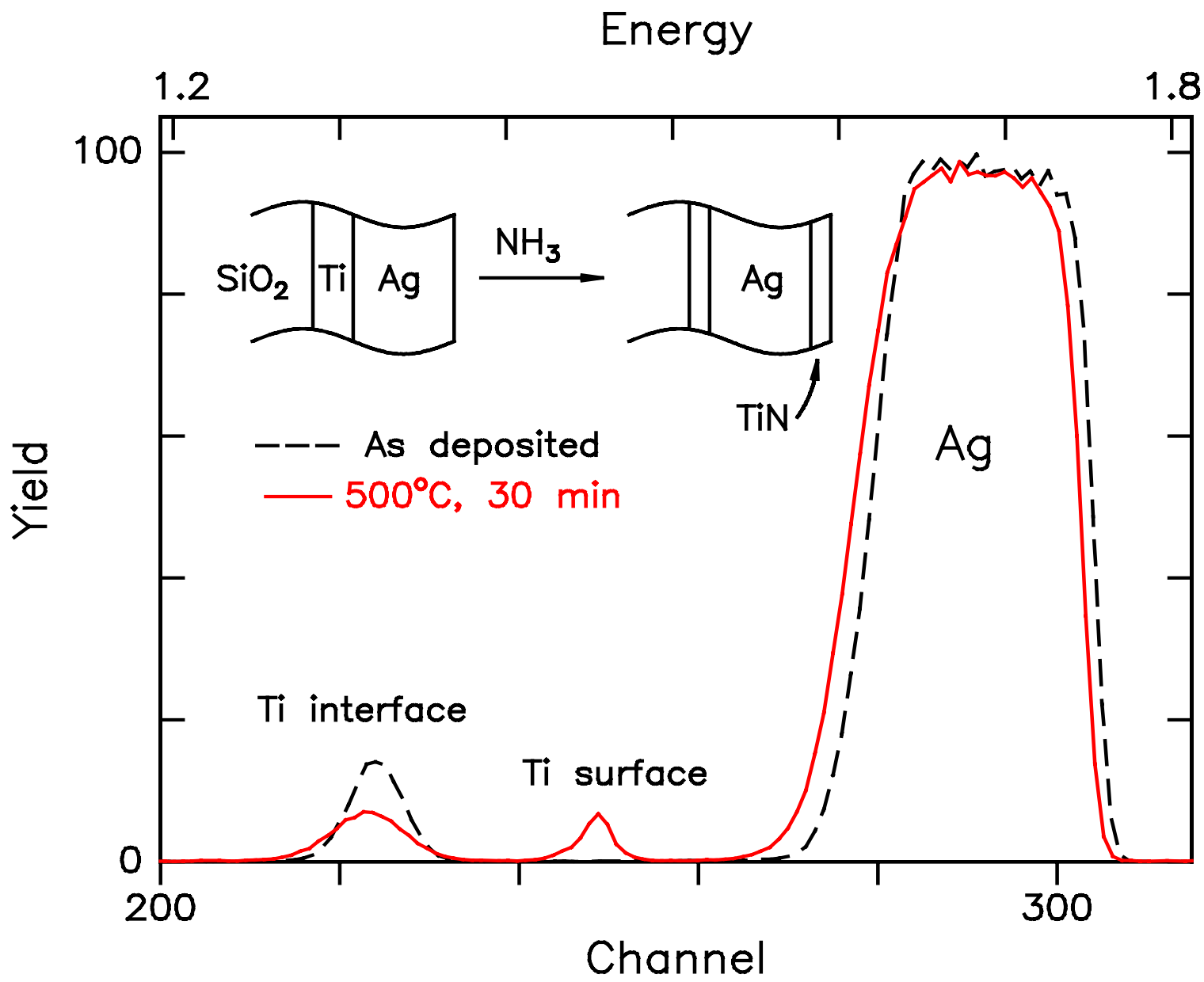
PHASE I: Blanket Thin Ag Films

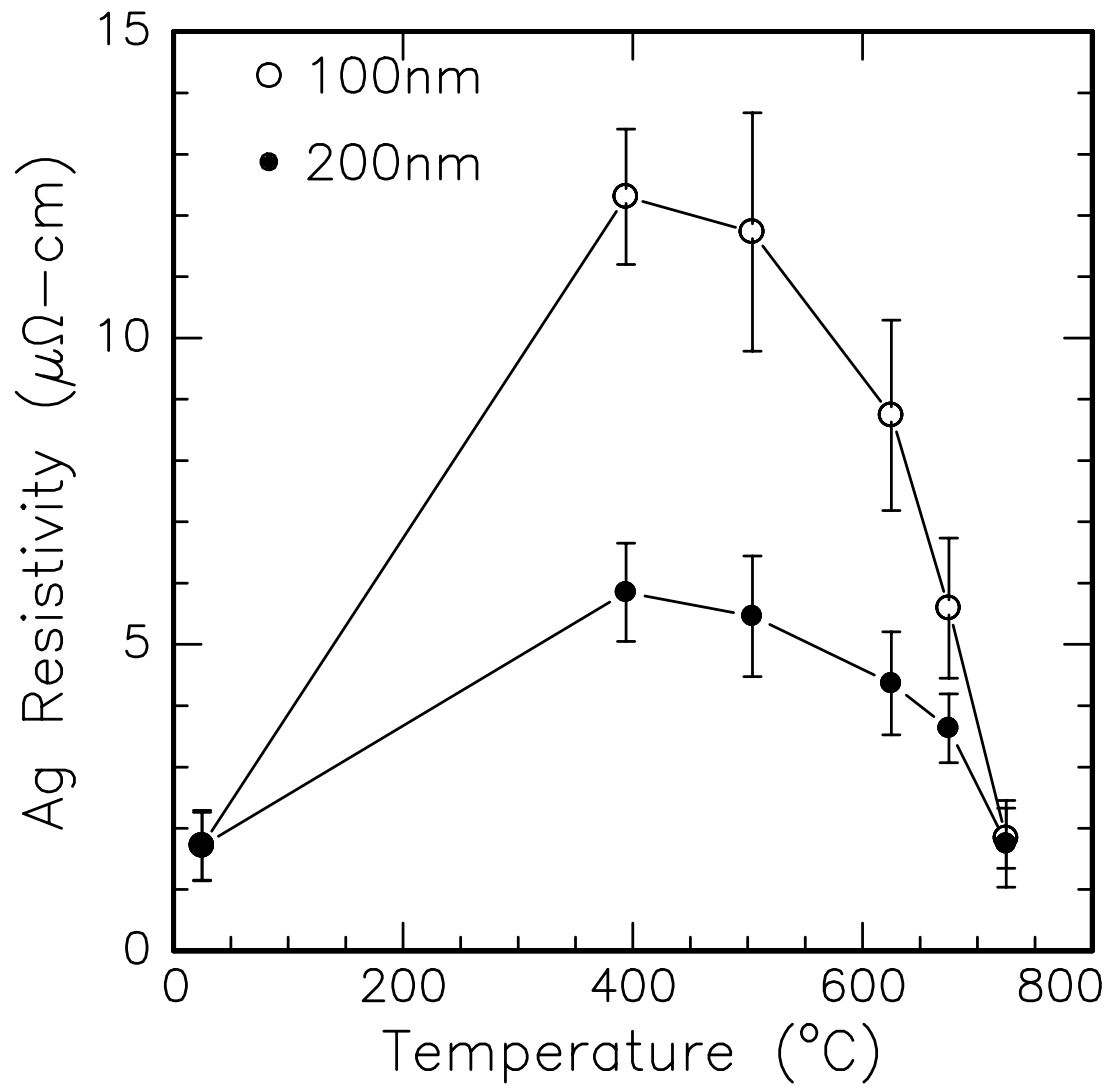


Encapsulation
process

Texture and
microstructure
optimization

Resistivity
measurement





$$\rho_{alloy} = \rho_0 + \kappa c_i \quad \kappa = \frac{\Delta\rho}{\Delta c_i}$$

	κ	
	Experimental	*Theoretical
Cu (% Ti)	13	16
Ag (% Ti)	1.8	NA
Ag (% Al)	1.3	1.95

*K. Schroeder, Handbook of Electrical Resistivities of Binary Metallic Alloys, CRC Press, 1983.

Encapsulation Process

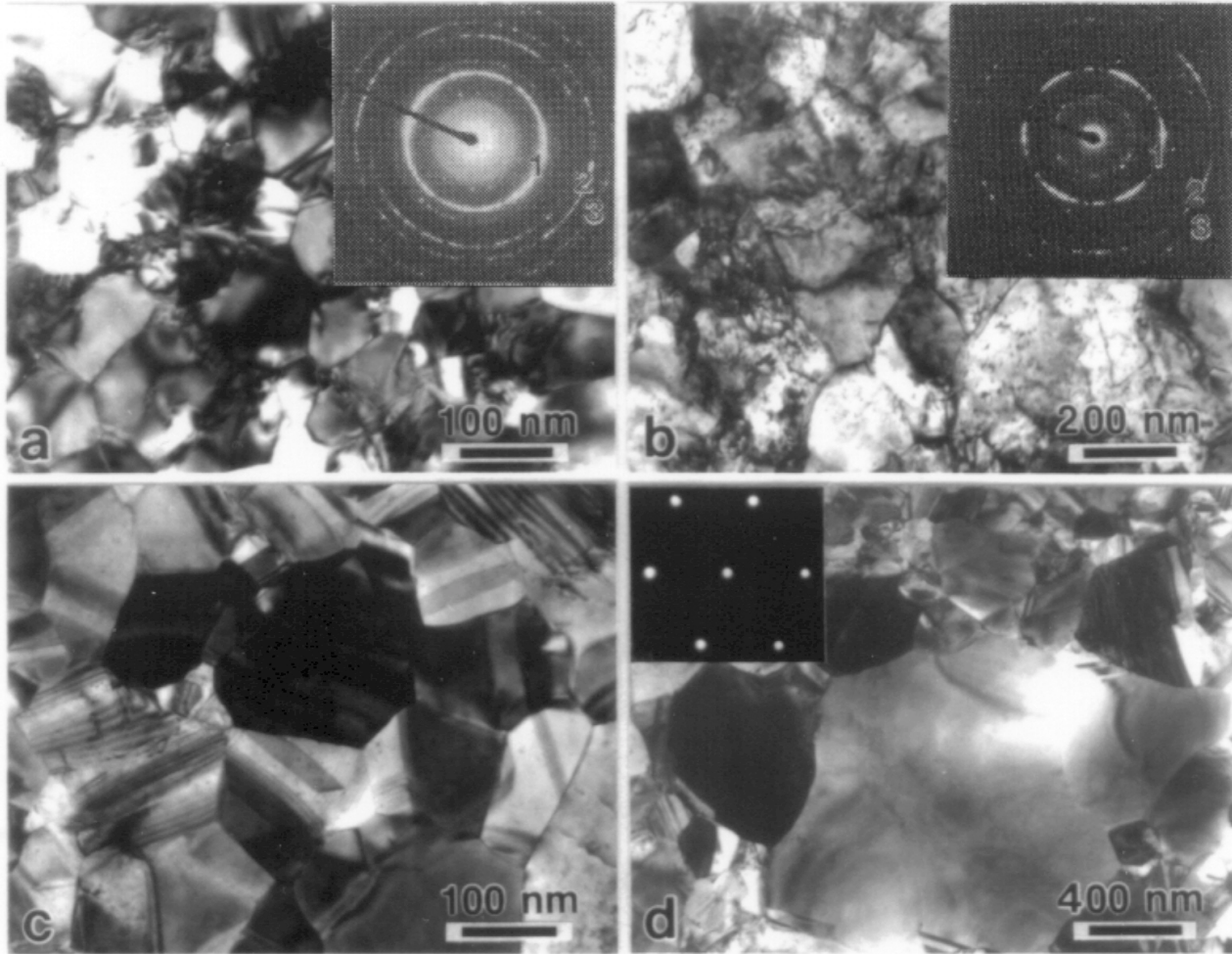
- **Ti or Al diffuses through Ag**
 - **Metal-N(O) forms on Ag surface**
 - **Metal reduces the underlying SiO_2**
 - **Negligible residual metal in Ag films**
- Ti and Al are ideal for encapsulation***

comparison of major texture parameters of Ag, Cu, and Al(Cu) thin films

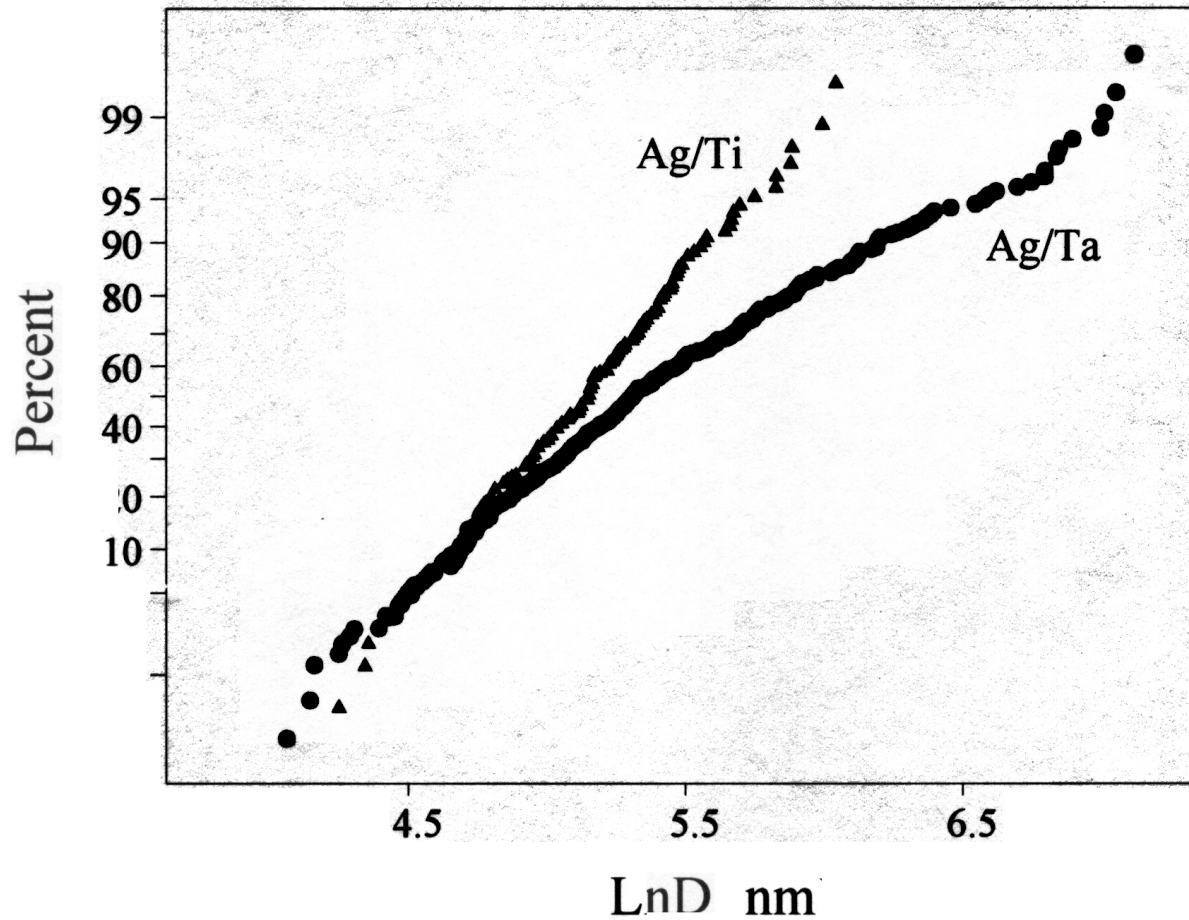
Ag	Condition	Volume fraction of grains <111> + <511> Random		FWHM (°) of <111> texture component
	As-deposited	0.74 + 0.17	0.09	5.6
	400 °C 1 hr	0.93 + 0.05	0.02	3.0

Cu	Condition	Volume fraction of grains <111> + twin <200> random			FWHM (°) of <111> texture component
	As-deposited	0.63 + 0.13	0.10	0.14	28.4
	400 °C 1 hr	0.74 + 0.07	0.12	0.07	21.0

AlCu	Condition	Volume fraction of grains <111> Random		FWHM (°) of <111> texture component
	As-deposited	0.89	0.11	10.2
	400 °C 1 hr	0.91	0.09	6.6



a) as-deposited, and b) 600 °C 15 min annealed Ag/Ti bilayers, c) as-deposited, and d) 600 °C 15 min annealed Ag/Ta bilayers

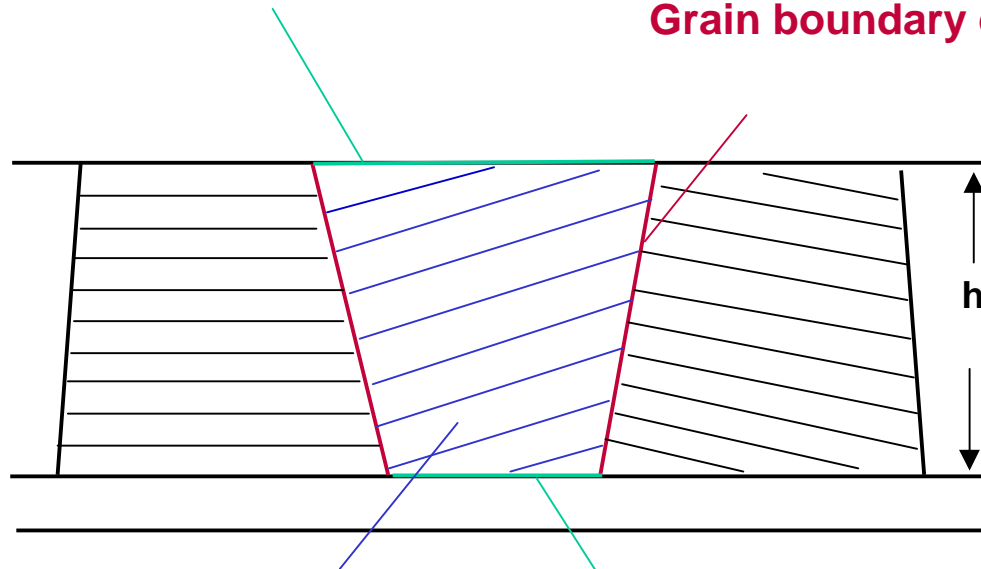


Grain size distribution of Ag films in Ag/Ti and Ag/Ta bilayers
Ag/Ti: Monomodal Ag/Ta: Bimodal

Surface and Strain Energy Controlled Grain Growth

Surface energy ($\Delta\gamma_s$)

Grain boundary energy (γ_{GB})



Strain energy (ΔF)

Interfacial energy ($\Delta\gamma_i$)

- ◆ **Driving force:** energy minimization
- ◆ **Grain growth rate:** $v = m[\Delta\gamma/h + \Delta F + \gamma_{GB}(1/r_a - 1/r)]$

Ag Texture and Microstructure

◆ Underlayer Dependence

- Ti: *Strong Ag <111> texture, uniform microstructure*
- Ta, Cr: *Random orientation, abnormal grain growth*

◆ Mechanisms

- *Lattice match*
- *Surface-and-strain-energy-controlled grain growth*